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**Peptide Formation on Layered Mineral Surfaces: The Key Role of Brucite-like Minerals on the Enhanced Formation of Alanine Dipeptides**

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# 1 Peptide Formation on Layered Mineral Surfaces: The Key Role of Brucite- 2 like Minerals on the Enhanced Formation of Alanine Dipeptides

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12  
13 Keywords : Layered double hydroxides, salt induced peptides formation, amino acids,  
14 Bernal's hypothesis, surface polymerization.

## 15 Abstract

16 Alkaline hydrothermal vent environments have gained much attention as potential sites for  
17 abiotic synthesis of a range of organic molecules. However the key process of peptide  
18 formation has generally been undertaken at lower pH, and using dissolved copper ions to  
19 enhance selectivity and reactivity. Here, we explore whether layered precipitate minerals,  
20 abundant at alkaline hydrothermal systems, can promote peptide bond formation for surface-  
21 bound alanine under cycles of wetting and drying. While we find low level activity in brucite  
22 and binary layered double hydroxide carbonate minerals (typically < 0.1% yield), the  
23 inclusion of structural copper to form a ternary layered double hydroxide mineral significantly  
24 increased the yield to > 7 %. However the performance decreased over successive  
25 wetting/drying cycles. Control experiments show that this high degree of dipeptide formation  
26 cannot be attributed to leached copper from the mineral structure. While only dipeptides are

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3 27 observed, the yields obtained suggest that such processes, if occurring on the early Earth,  
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5 28 could have added to the pool of available biological building units.  
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## I. Introduction

One of the prerequisite conditions for the emergence of life on Earth by the formation of more complex systems from simpler organic precursors, would have been the presence of a stable liquid hydrosphere in which simple endogenic or exogenic organic compounds may interact to form more complex systems. Along with the formation of fatty acids, sugars and other protobiomolecules, a key step in such hypothesised processes is the formation of short chained peptides from simple amino acids <sup>1</sup>.

While genetic information is carried and transferred by deoxyribonucleic acid (DNA) and ribonucleic acid (RNA), RNA may also act as a catalyst for biochemical reactions including autocatalysis. This led to the general proposal that life could have started using RNA –the so-called RNA World hypothesis <sup>2,3</sup>. Although activated RNA nucleobases can be polymerized in the presence of montmorillonite <sup>4-6</sup>, and RNA nucleotides are able to self-organize <sup>7</sup>, RNA molecules themselves are relatively unstable <sup>8</sup> and only a few long RNA sequences may exhibit catalytic activity <sup>9</sup>. From a geochemical perspective, the formation of nucleobases, nucleosides and nucleotides involves concentrations of chemicals that are difficult to achieve in terms of our present understanding of likely prebiotic Earth chemistry <sup>10,11</sup>. Unlike RNA, amino acids are known to form readily in experiments replicating putative reducing early Earth atmospheres <sup>12</sup> and are present in significant concentrations in carbonaceous chondrite meteorites <sup>13, 14</sup>, indicating exogenic synthesis as well. Moreover, peptides and proteins exhibit enhanced catalytic properties and stability relative to RNA <sup>15</sup>. The formation of peptides from amino acids has already been demonstrated by Rode and co-workers using the salt-induced peptide formation (SIPF) reactions <sup>16</sup>. In SIPF, the amino acids condense in the presence of concentrated sodium chloride solutions and form peptide bonds to yield di-peptides and oligopeptides. The presence of dissolved copper cations leads to increased yields and selectivity <sup>17</sup>. The SIPF reactions exhibit interesting properties in a pre-

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3 55 biological context, including regio-selectivity toward *alpha*-amino acids, and also  
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5 56 enantioselectivity toward the L-forms of some amino acids <sup>17-19</sup>. Although glycine can  
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7 57 polymerise up to Gly<sub>6</sub>, the SIPF reaction usually forms only small peptides, generally limited  
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9 58 to AA<sub>2</sub> or AA<sub>3</sub> where AA is an amino acid. However, on the early Earth, these small peptides  
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11 59 may have undergone further evolution pathways via chain elongation and stabilisation on  
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13 60 mineral surfaces <sup>20</sup>. Peptide formation can also be catalysed on mineral surfaces or edges as  
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15 61 illustrated on montmorillonite, kaolinite and Fe(II) rich smectite <sup>20-24</sup>. However, the yield is  
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17 62 low (typically less than 1% reaction yields for 5 wetting/drying cycles), and these reactions  
18  
19 63 appear unlikely as a major contributor to the prebiotic formation of peptides from amino  
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21 64 acids. Aside the wetting/drying cycles as driving force for peptide formation, thermal  
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23 65 activation may also be suggested as an complementary way for producing peptides. Studies  
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25 66 reported peptide formation over silica, Fe-oxides and metal-oxides with yield reaching 10-  
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27 67 20% in some cases at relatively low thermal activation (60-120°C)<sup>25-27</sup>. Although less relevant  
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29 68 in the context of prebiotic chemistry, high temperature thermal copolymerization or vapor  
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31 69 phase amino-acid deposition on SiO<sub>2</sub> and TiO<sub>2</sub> have produced longer peptides that presents  
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33 70 secondary structure, that may be considered as a crucial step in the path leading to living cell  
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35 71 <sup>28, 29</sup>.

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39 72 The SIPF reaction remains a possible mechanism although more investigations are  
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41 73 needed to improve the peptide yield and the enantioselectivity under plausible prebiotic  
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43 74 conditions. While copper is one of the most abundant transition metal ions in sea water <sup>30</sup>, the  
44  
45 75 anoxic state and estimated pH of the early ocean makes the availability of free Cu<sup>2+</sup> more  
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47 76 questionable <sup>31, 32</sup>, and the concentrations required for SIPF are usually relatively high.  
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49 77 Copper is readily removed from seawater, being sequestered by chelation, precipitation or  
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51 78 adsorption onto other mineral surfaces. To achieve the required high salt concentration to  
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53 79 drive condensation reactions and peptide bond formation, the SIPF reaction has been  
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3 80 postulated to have taken place in evaporation cycles in coastal lagoons or salt lakes, and  
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5 81 subsequent wetting could have been achieved by incoming high tides, waves or rain in order  
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7 82 to start the reaction again. Experimentally, simulated evaporation/wetting cycles achieve a  
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9 83 higher rate of peptide bond formation than constant volume experiments with reflux cooler to  
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11 84 prevent evaporation<sup>33,34</sup>.

13 85 Since the discovery of abiotic hydrocarbons and rich microbial activity in oceanic  
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15 86 alkaline hydrothermal fields, the idea that life may have originated in alkaline surroundings  
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17 87 has gained credence<sup>35-38</sup>. Sedimentology records and evidence of nitrogen isotope  
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19 88 fractionation in the Tumbiana Formation may indicate that the occurrence of alkaline lakes  
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21 89 was frequent in the Archean environment<sup>39</sup>. Present day alkaline lakes and alkaline  
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23 90 hydrothermal vents are much more biologically productive than freshwater water systems,  
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25 91 and may have been so in the past<sup>40</sup>. Elevated pH conditions also induce the precipitation of  
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27 92  $\text{CaCO}_3$ , which limits  $\text{Ca}^{2+}$  binding to phosphate thus aiding the formation of phosphate based  
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29 93 polymers, as well as enhancing the polymerisation of HCN that can play a role in the abiotic  
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31 94 formation of amino acids<sup>41</sup>. Hanczyc *et al.* reported that the formation of vesicles is  
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33 95 facilitated in alkaline media, consistent with their work showing that the origin of life may  
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35 96 have been compartmentalized in lipid membrane<sup>42</sup>. However, alkaline conditions are not  
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37 97 compatible with the peptide- forming SIPF reactions. SIPF reactions require dissolved  $\text{Cu}^{2+}$   
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39 98 ions, whereas  $\text{Cu}^{2+}$  precipitates if the water pH is alkaline, or the water oxic, depending on  
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41 99 temperature. Thus, in order to form peptide bonds abiotically, and under alkaline conditions, a  
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43 100 different mechanism to that of the homogeneous SIPF reactions is required.

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48 101 Among minerals formed in alkaline lakes and off-axis alkaline hydrothermal vents,  
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50 102  $\text{Mg}(\text{OH})_2$ , i.e. brucite, can be abundant<sup>43, 44</sup>. The structure of brucite  $\text{Mg}(\text{OH})_2$  consists of  
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52 103 magnesium ions octahedrally coordinated to hydroxide ions. These octahedral units form  
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54 104 infinite edge-sharing layers with the hydroxide ions sitting perpendicular to the plane of

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3 105 cations. The layers are stacked on top of one another to form a three dimensional structure <sup>45</sup>.  
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5 106 The local geometry around the metal and the hydroxide ions are distorted. This distortion has  
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7 107 been discussed from a molecular orbital point of view, and, based on unit cell parameters and  
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9 108 bond distances, it has been suggested that all brucite-like layered structures, including layered  
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11 109 double hydroxides, exhibit similar distortion, <sup>46, 47</sup>. Brucite can readily undergo mineral  
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13 110 transformations through isomorphous substitution of  $Mg^{2+}$  by  $M^{3+}$  species during  
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15 111 precipitation to form layered double hydroxides (LDHs), a family of layered minerals with  
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17 112 high layer cationic charge <sup>48, 49</sup>. Moreover, LDHs would have formed on the early Earth as a  
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19 113 result of seafloor weathering processes such as serpentinization, or by reaction with solutions  
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21 114 with high salinities <sup>50, 51</sup>. Other cations such as Mn(II), Al(III), Fe(III), Fe(II), Ni(II) or Cu(II)  
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23 115 may also be incorporated within the brucite-like structure <sup>52</sup> and LDHs are notable for their  
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25 116 anion exchange and concentrating capacity <sup>53</sup>. The main group of LDH materials, as used  
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27 117 here, are thus described by the formula:  $[M^{II}_{1-x}M^{III}_x(OH)_2] (A^{n-})_{x/n} \cdot mH_2O$ , where  $M^{II}$  and  $M^{III}$   
28  
29 118 are di- and trivalent cations of similar radii, respectively,  $A^{n-}$  is an exchangeable interlayer  
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31 119 anion with charge n, and x represents the  $M^{II}/(M^{III}+M^{II})$  molar fraction.  
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35 120 The present study explores the possible role that may have been played by these  
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37 121 layered hydroxide minerals in early Earth chemistry, leading to the formation of peptides.  
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39 122 While the SIPF reaction required acidic conditions, the experimental conditions applied in this  
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41 123 study are alkaline with pH values of 9-9.5, and, in particular, include experiments conducted  
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43 124 even in the absence of dissolved  $Cu^{2+}$  species. To achieve textural and structural  
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45 125 homogeneity, we used brucite starting materials that were iso-structurally modified to form  
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47 126 various layered double hydroxides with different compositions. The materials studied here are  
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49 127 all based on commonly shared octahedrally coordinated cation sheets, and the results  
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51 128 presented may thus possibly be extended to other minerals sharing similar structures. Peptide  
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53 129 formation was followed in all cases using alanine monomers, on the basis of data from  
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3 130 molecular simulations reported in our previous work suggesting preferred reactivity of this  
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5 131 amino acid at layered mineral interfaces<sup>54</sup>.

## 8 132 **2. Material and methods**

### 11 133 **2.1. Reagents**

14 134 The cation providing salts (MgCl<sub>2</sub>·6H<sub>2</sub>O, AlCl<sub>3</sub>·6H<sub>2</sub>O and CuCl<sub>2</sub>·2H<sub>2</sub>O) and the carbonate  
15  
16 135 (Na<sub>2</sub>CO<sub>3</sub>) used for the preparation of the LDH and brucite phases were supplied by Sigma  
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18 136 Aldrich and used as received. L-Alanine and D-Alanine amino acids were supplied by Sigma  
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20 137 Aldrich and the dipeptide standard L-Ala-Ala was supplied by Bachem AG, Switzerland. An  
21  
22 138 achiral high pressure liquid chromatography (HPLC) column was used so that no D-Ala-Ala  
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24  
25 139 was required. The ion-pairing reagent, sodium *n*-hexanesulfonate monohydrate, NaCl,  
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27 140 KH<sub>2</sub>PO<sub>4</sub> and concentrated phosphoric acid (50 wt.%) were also provided by Sigma Aldrich.

### 30 141 **2.2. Synthesis of mineral phases**

33 142 Synthesis of brucite: A 1 M NaOH solution was added dropwise (0.3 mL/min) to 50 mL of a  
34  
35 143 0.4 M solution of MgCl<sub>2</sub> dissolved in water. The reaction was stopped when the pH of the  
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37 144 solution reached pH 12. The solution was then allowed to stir for 2 hours. Un-reacted salts  
38  
39 145 were removed from the solution by multiple washing/centrifugation steps.

42 146 Synthesis of layered double hydroxides: Layered double hydroxides (LDHs) were prepared by  
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44 147 the precipitation at low supersaturation method. The co-precipitation reaction was performed  
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46 148 by the constant addition (0.3 mL/min) of a 0.4 M solution containing the divalent (Mg<sup>II</sup>, Zn<sup>II</sup>,  
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48 149 Cu<sup>II</sup>, Ni<sup>II</sup>) and trivalent cations (Al<sup>III</sup>) in the desired ratio into a solution of 0.25 M Na<sub>2</sub>CO<sub>3</sub>.  
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50 150 The pH was maintained at 10 for all preparations by the simultaneous addition of a 1 M  
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52 151 NaOH solution. Once the co-precipitation reaction was achieved, the slurries were vigorously

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3 152 stirred for 2 hours. Then, the solution was centrifuged and thoroughly washed with water.  
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5 153 The slurries were then dried in an oven at 60 °C overnight.  
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### 8 154 **2.3. Salt induced peptide formation experiments**

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11 155 To allow comparison of the efficiency of LDH minerals for peptide bond formation, a set of  
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13 156 controls using homogeneous SIPF reactions was first performed. Concentrations of alanine,  
14  
15 157 copper and chloride were 100 mM, 50 mM and 400 mM, respectively. According to previous  
16  
17 158 work a ratio AA:Cu of 2:1 was the most effective for optimising the SIPF reaction efficiency  
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19 159 [52]. Experiments were performed over a period of 14 days, corresponding to 14  
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21 160 wetting/drying cycles. At selected intervals, solutions were sampled, analysed by HPLC, and  
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23 161 the quantity of di-alanine peptides formed was measured. Both L-Alanine and D-Alanine  
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25 162 were used as starting reactants, in order to measure possible enantioselectivity.  
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### 30 163 **2.4. Evaporation cycle experiments**

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32 164 **Homogeneous SIPF experiments** The formation of peptide bonds was carried out in drying  
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34 165 and wetting cycle experiments that simulate the environment in tidal lagoons on a primordial  
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36 166 Earth. A classical SIPF experiment was first performed serving as a reference to assess the  
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38 167 influence of LDH chemistry on peptide yield formation and enantioselectivity. The SIPF  
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40 168 solution consisting of 500 mM NaCl and 40 mM CuCl<sub>2</sub> was used with a starting solution of  
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42 169 100 mM of L or D Alanine. After mixing, 1 mL aliquots of the reaction solution were  
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44 170 transferred into 2 mL Eppendorf® and then held in a thermostatic oven at 80°C to start the  
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46 171 first evaporation cycle within 24h. After each cycle, 1 mL of ultrapure water was added to the  
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48 172 residue for the next cycle under the same conditions. After 0, 1, 4, 7 and 14 cycles, the  
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50 173 evaporated sample were removed and frozen dry at -20°C for further analyses.  
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54 174 **Reactions in the presence of brucite and LDH minerals** In order to investigate the role of the  
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56 175 chemical composition of LDHs involved in peptide formation and the possibility that minerals  
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3 176 can advantageously replace  $\text{Cu}^{2+}$ , a very similar protocole as that of the SIPF reaction was  
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5 177 adopted. An aliquot of 1 mL of a solution containing 500 mM NaCl and 100 mM of L or D  
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7 178 Alanine was contacted with 0.45 mmol of the materials investigated. Since the chemical  
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9 179 composition of the solid phases was the studied parameter, the mass of the materials was  
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11 180 adjusted to fit the required molar quantity. This allows the efficiency of the catalyst to be  
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13 181 directly compared through the observed conversion. The conditions of the wetting-and-drying  
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15 182 cycle was identical as that of the SIPF reaction described above. For the sake of completeness,  
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17 183 one experiment was carried out by contacted the SIPF reagents with LDH.

#### 184 **2.4. Characterisation methods**

185 Powder X-ray diffraction (XRD). Patterns were recorded with a Phillips X'Pert Pro MPD  
186 diffractometer at Oxford University in reflection geometry using Cu  $\text{K}\alpha 1$  radiation ( $\lambda =$   
187  $1.5406 \text{ \AA}$ ). Samples were finely ground and continuously rotated to improve statistics. The  $2\theta$   
188 range was between  $5\text{--}70^\circ$  with a step size of  $0.01671^\circ$  and a scan time per step set at 2 s  
189 giving a scan rate of  $2^\circ/\text{min}$ . The cell parameters were determined from peak profile analysis  
190 using pseudo-Voigt functions<sup>55</sup>.

191 Raman spectroscopy. The Raman analyses were performed on a Jobin Yvon T64000 equipped  
192 with nitrogen cooled Charged Coupled Device detector and a motorized XYZ stage in the  
193 University of Oxford. The samples were placed on a glass holder and mounted in the focal  
194 plane of an Olympus X50 objective (N.A = 0.55). The spot area was around  $2 \mu\text{m}^2$ . A  $514.53$   
195 nm exciting radiation was used and the power was adjusted depending on the nature of the  
196 samples to prevent sample degradation. The spectral resolution was about  $4 \text{ cm}^{-1}$  and the  
197 precision on the wavenumber was lower than  $1 \text{ cm}^{-1}$ .

198 High performance liquid chromatography analysis. After a certain number of cycles, the  
199 residues were dissolved again, filtered and analysed by reverse-phase ion-pairing  
200 chromatography on a Waters 600s system, equipped with a diode array detector. Quantitative

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3 201 and qualitative results were obtained by comparing the retention time, the response factor and  
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5 202 the UV spectra of standard alanine references. Depending on the quantity of dipeptide in the  
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7 203 solution, 2  $\mu\text{L}$  to 10  $\mu\text{L}$  of the sample solutions were injected into a Waters ODS column (5  
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9 204  $\mu\text{m}$ , 4.6 x 250 mm) equipped with a 20 mm pre-column of the same materials. The analyses  
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11 205 were performed in isocratic mode with a mobile phase containing 50 mM  $\text{KH}_2\text{PO}_4$  and 7.5  
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13 206 mM  $n\text{-C}_6\text{H}_{13}\text{SO}_3\text{Na}$  in ultrapure water, adjusted to pH 2.3 with concentrated  $\text{H}_3\text{PO}_4$ . The  
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15 207 UV/Vis detector was set at 200 nm. Two injections per run were performed.  
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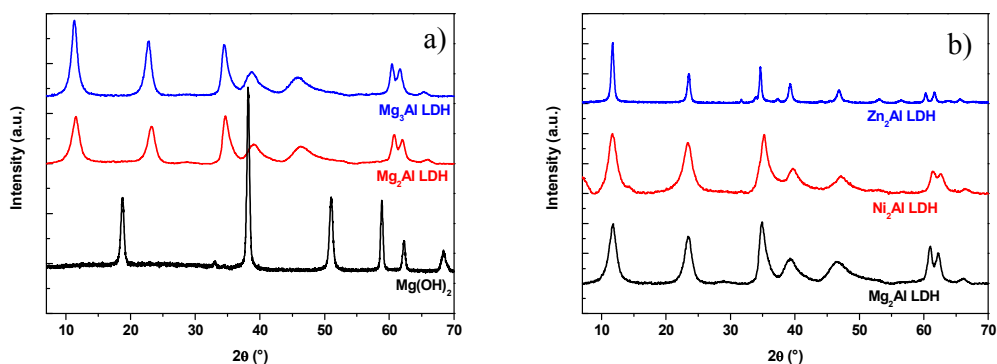
### 19 208 **3. Results and Discussion**

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21 209 In this section, we first report and discuss the structure of the starting mineral phases  
22  
23 210 prepared. The simple hydroxide ( $\text{Mg}(\text{OH})_2$ ), binary layered double hydroxides (Mg/Al, Ni/Al  
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25 211 and Zn/Al) and ternary layered double hydroxides (Mg/Cu/Al) are described in turn. Though  
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27 212 care was taken to obtain materials of similar crystallinity to ensure reasonable comparison  
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29 213 between peptide forming reactions, as shall be seen, variations were noted as a function of  
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31 214 composition of the layered material. Having understood the variation in mineral structure, the  
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33 215 activity of each system with respect to peptide bond formation is then explored.  
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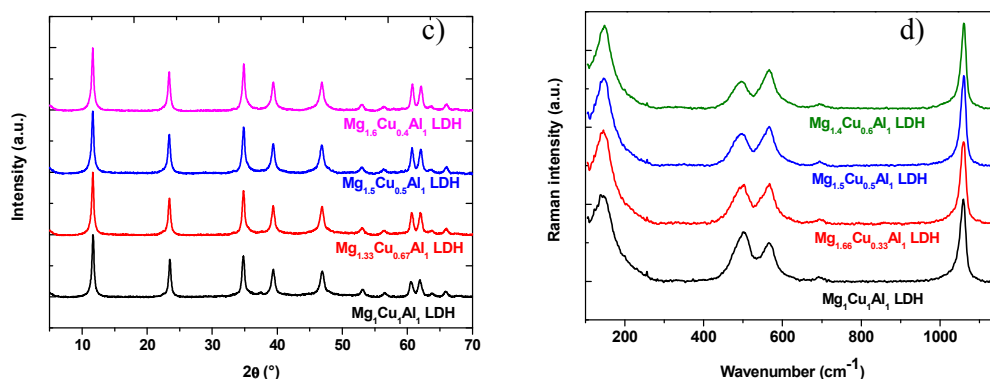
#### 37 216 **3.1. Structural characterisation of mineral phases**

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39 217 Simple and binary phases - brucite to layered double hydroxides. The XRD pattern of brucite  
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41 218 ( $\text{Mg}(\text{OH})_2$ ) is displayed in **Figure 1a** and indexed with a hexagonal symmetry (P-3m1) space  
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43 219 group. The cell parameters were determined as  $a = 3.140 \text{ \AA}$  and  $c = 4.732 \text{ \AA}$ , matching  
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45 220 reported cell parameters<sup>45</sup>. Carbonate was chosen as the LDH interlayer anion, as the most  
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47 221 plausible in alkaline water<sup>56</sup> and  $\text{Al}^{3+}$  is the trivalent cation. **Figure 1a** also displays the XRD  
48  
49 222 pattern of  $\text{Mg}_2\text{Al}$  and  $\text{Mg}_3\text{Al}$  LDH phases to identify the role of Mg content on the reactivity.  
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51 223 This comparison is enabled by the fact that the presence of carbonate anions prevents any  
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53 224 interlayer reactivity so that only surface sites are accessible. Lowering the layer charge  
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3 225 density (lower  $\text{Al}^{3+}$  content) leads to an increase in the interlayer spacing (shift toward lower  
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5 226 diffraction angle) as a result of the decrease of the electrostatic interaction between the  
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7 227 positive charge in the layer and the negative charge in the interlayer domain. The  $a$  parameter  
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9 228 also increases with decreasing layer charge density. Since the cell parameters evolve linearly  
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11 229 with the layer charge density, these two parameters  $a$  and  $c$  can be good indicators of the  
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13 230 successful control of the cation ratio within the LDH layers<sup>57</sup>. Different divalent cations were  
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15 231 also considered including  $\text{Mg}^{2+}$ ,  $\text{Ni}^{2+}$ , and  $\text{Zn}^{2+}$ ; all these cations being present in the early  
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17 232 ocean<sup>58</sup>. XRD patterns of the  $\text{M}^{\text{II}}\text{-Al}^{\text{III}}$  LDH phases synthesised are presented in **Figure 1b**.  
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19 233 Since the synthesised LDHs were prepared with a unique layer charge density and interlayer  
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21 234 anion, the  $c$  parameter ( $c = 3/2(d_{003} + 2d_{006})$ ) does not change (as shown in **Table 1**). The  
22  
23 235 position of the (110) XRD peaks enables calculation of the  $a$  parameter, which can be  
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25 236 correlated with the cation-cation distance. This parameter is cation dependent, varying with  
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27 237 the ionic radii of the cations at a given cation ratio<sup>59</sup>.  
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**Figure 1.** Powder X-ray diffraction patterns for minerals prepared in this study: (a) comparing the effect of Mg content in layered minerals. (b) comparing the effect of the nature of the divalent  $M^{II} = Mg^{II}, Ni^{II}, Zn^{II}$  in layered double hydroxides (c) Ternary layered double hydroxides with  $M^{II}:M^{III} = 2$  and different  $Cu^{II}$  content. The corresponding Raman spectra of these ternary phases are displayed in (d).

Although similar synthesis methods were used, there is a variation in crystallinity that depends on the cationic composition and stoichiometry. The apparent domain size, calculated with the Scherrer equation ( $K = 1$ ), along the [001] direction corresponds to the average thickness of the crystallites, while the apparent domain size along the [110] direction allows the apparent diameter of LDH crystals to be estimated. The domain size along the [110] direction is higher than that along the [001] direction as a result of the classical platelet-like morphology observed for LDH particles prepared by co-precipitation<sup>57</sup> (**Table 1 and Table ESI 1**). Interestingly, the nature of the cations dictates the crystallinity of the LDH phases. Among all the cation couples used,  $Zn_2Al$  exhibits much larger coherent domain size in both directions. This enables investigation of whether the cations or the textural properties play the major role in the reactivity of these phases.

*Ternary Layered Double Hydroxides* To investigate the possible role of incorporating Cu in LDHs, ternary Mg/Cu/Al LDH materials were also prepared. **Figure 1c** shows the XRD

259 patterns of the ternary  $\text{Mg}_{2-y}\text{Cu}_y\text{Al}$  LDH phases. All XRD peaks can be indexed in the  
 260 common R-3m group of LDH, indicating the samples are phase pure. The symmetry of the  
 261 XRD lines suggests a regular stacking sequence, as expected when carbonate is intercalated  
 262<sup>60</sup>. The positions of the (001) and (110) XRD lines were used to calculate the cell  
 263 crystallographic parameters. Since the XRD lines exhibit quite uniform broadening, the  
 264 apparent diameter along the [001] direction and along the [100] direction can also be  
 265 calculated<sup>61</sup>. These values are reported in **Table 1**.

266 Variation of the cell parameters with changing  $\text{Cu}^{2+}$  content gives direct evidence of  
 267 the incorporation of the copper ions within the brucite-like sheet. The overall layer charge  
 268 density of the ternary system is constant and the *c* parameter did not vary. In contrast, the unit  
 269 cell *a* parameter increased with increasing copper content through the replacement of smaller  
 270  $\text{Mg}^{2+}$  ions by larger  $\text{Cu}^{2+}$  ions. From the domain size indicated by the Scherrer equation, it can  
 271 be seen that the  $\text{Cu}^{2+}$  content has a negligible effect on the apparent domain size along the  
 272 [001] direction, i.e. thickness of crystallite, but a significant effect on the apparent domain size  
 273 along the [110] direction i.e. the basal plane of the crystallites. This may be explained by  
 274 octahedral Jahn-Teller distortion related to the presence of octahedral  $\text{Cu}^{2+}$  ions. The coherent  
 275 domain size decreases as the  $\text{Cu}^{2+}$  content increases.

276 **Table 1.** Cell parameters and calculated domain sizes for the prepared ternary Mg/Cu/Al layered double  
 277 hydroxide minerals. The error on the determination of cell parameters was estimated at 0.03%.

LDH	Cell parameters (Å)		Apparent domain size (nm)	
	<i>c</i> (Å)	<i>a</i> (Å)	along [001]	along [110]
$\text{Mg}_1\text{Cu}_1\text{Al}_1$ LDH	22.824	3.054	22.8	33.1
$\text{Mg}_{1.33}\text{Cu}_{0.66}\text{Al}_1$ LDH	22.819	3.050	22.2	42.7
$\text{Mg}_{1.5}\text{Cu}_{0.5}\text{Al}_1$ LDH	22.823	3.048	21.5	46.2
$\text{Mg}_{1.6}\text{Cu}_{0.4}\text{Al}_1$	22.824	3.046	21.3	48.8

Mg <sub>2</sub> Al	22.684	3.037	7.9	23.6
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278

279 To ensure that Cu<sup>2+</sup> had been incorporated within the LDH layer, and not precipitated as  
280 another phase, Raman spectra were obtained (**Figure 1d**). Raman spectroscopy probes local  
281 structure, rather than the long-range structure characterised by XRD, and has the advantage of  
282 being able to detect even poorly crystallised phases. The Raman bands located at 1056 cm<sup>-1</sup>  
283 and 680 cm<sup>-1</sup> correspond to the symmetric stretching and bending modes of the carbonate  
284 anion respectively. At lower wavenumbers, an intense band at around 120 cm<sup>-1</sup> is attributed  
285 to translational mode of the interlayer species<sup>62</sup>. The two bands situated at 500 cm<sup>-1</sup> and 560  
286 cm<sup>-1</sup> are assigned to the LDH phase as reported elsewhere<sup>63</sup>. No other band is observed in  
287 this spectral range, confirming that the LDH prepared are phase pure and no second Cu<sup>2+</sup>  
288 phase is present.

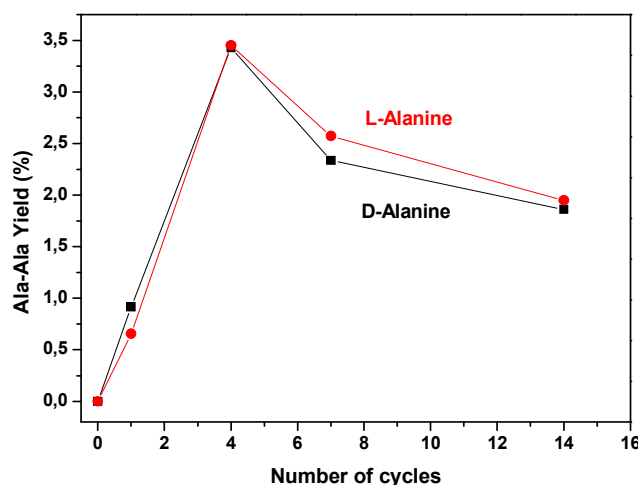
289 Interestingly, the positions of the LDH bands are the same even when the proportion  
290 of Cu<sup>2+</sup> in the sheet is varied, indicating that the intralayer bonding is similar in all cases.  
291 However, the relative intensities of the two bands present at 560 cm<sup>-1</sup> and 500 cm<sup>-1</sup> show  
292 systematic changes reflecting the content of the Cu<sup>2+</sup> ions (**Figure ESI 2**). By analogy with  
293 binary Mg<sub>2</sub>Al LDHs, where the most intense band is located at 560 cm<sup>-1</sup>, the Raman band  
294 situated at 560 cm<sup>-1</sup> is attributed to the Mg<sub>2</sub>AlOH domain, with the one at 500 cm<sup>-1</sup> assigned  
295 to the Cu<sub>2</sub>AlOH domain, suggesting cations in the ternary systems may form distinct  
296 domains. A similar interpretation was proposed by Leroux *et al.* on systems of the type  
297 CoCuAl on the basis of EXAFS spectroscopy<sup>64</sup>.

### 298 **3.2 Salt induced peptide formation experiments**

299 The results obtained for the control experiments with no LDH present, and using salt induced  
300 peptide formation (SIPF), are shown in **Figure 2**. The conversion yields are expressed as  
301 percentages of the initial amino acid concentration. Both curves are very similar, irrespective



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2  
3 302 of the chirality of the amino acid. A net increase is observed until the fourth cycle, and then  
4  
5 303 the quantity of peptide formed decreases with increasing number of cycles. This may reflect  
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7 304 formation of dipeptides followed by hydrolysis, degradation or competition of alanine with  
8  
9 305 the dipeptide for the catalyst site. Formation of higher molecular weight peptides by  
10  
11 306 oligomerisation reactions of the dipeptides has not been observed in these systems. Any  
12  
13 307 enantioselectivity of the SIPF reaction is not obvious under our experimental conditions. Rode  
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15 308 and Suwannachot reported that the initial concentration of amino acid determines the extent of  
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17 309 enantioselectivity<sup>65</sup>. A slight enantioselectivity toward the L form is observed after the first  
18  
19 310 four cycles, as the dipeptide yield decreases, but the statistical significance of this result is  
20  
21 311 questionable. These SIPF syntheses were repeated with different cation configurations.  
22  
23 312 Repetition of the simple SIPF reaction after removing either Cu<sup>2+</sup> or NaCl, or on changing the  
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25 313 nature of the divalent cation (Ni<sup>2+</sup>, Zn<sup>2+</sup> or Mg<sup>2+</sup>) resulted in no detectable amounts of di-  
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27 314 peptides.  
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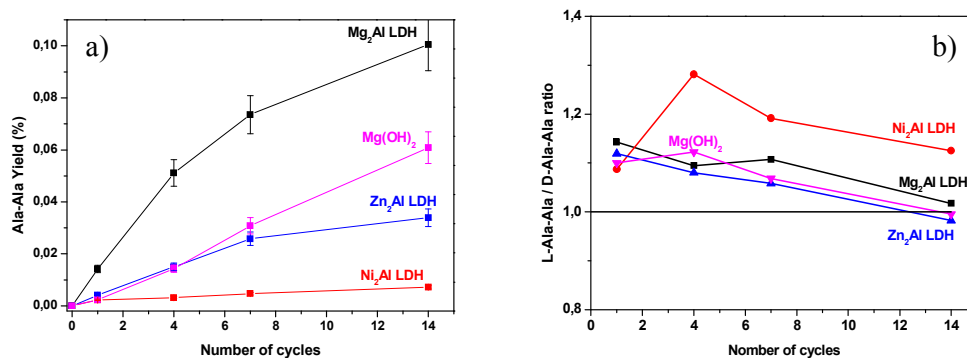
316  
317 **Figure 2.** Salt induced peptide formation reaction yields of di-alanine (L and D) is plotted versus number of  
318 wetting/drying cycles.

### 319 **3.3 Mineral induced peptide formation experiments**

320 Brucite and binary layered double hydroxide minerals The results of peptide forming  
321 experiments for different LDH compositions are shown in **Figure 3**. The four curves all show  
322 a continuous increase in peptide formation over the entire number of wetting/drying cycles in  
323 contrast to the behaviour observed in the homogeneous SIPF reaction (**Figure 2**). This  
324 suggests that the LDH surface stabilizes or protects dipeptides against subsequent hydrolysis  
325 more efficiently than the SIPF reaction. After 14 cycles, the peptide yield reaches 0.10% for  
326 Mg<sub>2</sub>Al LDH, 0.06% for Mg(OH)<sub>2</sub>, 0.03% for Zn<sub>2</sub>Al LDH and less than 0.01% for Ni<sub>2</sub>Al  
327 LDH. These values are much lower than those obtained from the homogeneous SIPF  
328 reactions. Note that the pH of the starting solution in contact with the solid catalyst was very  
329 similar in each case, because of the strong buffering capacity of these carbonate materials,  
330 with values ranging from pH 9 to pH 9.5. Based on the values of the apparent domain size  
331 determined by XRD (Figure ESI 1), it is inferred that the crystal properties of Ni<sub>2</sub>Al and  
332 Mg<sub>2</sub>Al LDH are similar, suggesting that the difference of reactivity cannot be simply  
333 attributed to the accessibility of surface active sites and that the critical parameter is the nature  
334 of the cation. Since carbonate was the charge balancing anion, no anion exchange reactions  
335 would have occurred owing to the stability of carbonate LDHs, and only the exterior surfaces  
336 should be accessible to species in solution. In addition, though edge cations may be exposed,  
337 the cations in LDHs are mainly located in the centres of hydroxide octahedra, and are  
338 therefore not readily accessible for bonding with solution amino acids. Interestingly, both Mg-  
339 based materials exhibit the higher conversion yield, showing that Mg-containing layers are the  
340 most reactive/stabilising for peptide formation.

341 The small conversion yield in all cases implies that most of the surface sites do not  
342 participate in the reaction, with only a small fraction of all surfaces sites being active, for  
343 example at structural defects or LDH particle edges. Interestingly, though the structures are

344 similar,  $\text{Mg}(\text{OH})_2$  and  $\text{Mg}_2\text{Al}$  LDHs exhibit substantially different reactivity for alanine  
 345 dipeptide formation. Assuming a surface mechanism, the only sites that are accessible from  
 346 solution are the hydroxide layers and, for the LDH, external bound carbonate anions. Peptide  
 347 bond formation requires reaction at surface basic sites and, in LDH systems, the strength of  
 348 the basic sites depends on the nature of the cations, particular in terms of electronegativity and  
 349 the electron density gradient in the layers influencing the basicity of the hydroxyl group. The  
 350 greater the electronegativity of the atom, the weaker the basic sites formed. In the LDH  
 351 structure, and assuming cation ordering within the layer, all OH groups are similar, and all are  
 352 bonded to two divalent cations and one trivalent cation ( $\text{Mg}_2\text{Al-OH}$  moieties)<sup>66</sup>. Since the  
 353 only trivalent cation present was  $\text{Al}^{3+}$ , comparison of the electronegativities of the divalent  
 354 cations between the various LDHs allows an estimation of the relative basicity of the different  
 355 surfaces. The electronegativity of the cations follows  $\text{Mg}$  (1.31) <  $\text{Zn}$  (1.65) <  $\text{Ni}$  (1.91).  
 356 These are in good accord with the respective peptide yields after 14 days. Among the LDHs  
 357 tested, Mg based compositions with the most basic sites also gave the highest yields as  
 358 expected.



359  
 360 **Figure 3.** Results from peptide forming reactions on brucite and binary layered double hydroxides, where  $\text{M}^{2+}$   
 361 was either Mg, Zn or Ni, showing: (a) yields of di-alanine against the number of wetting/drying cycles; (b) the  
 362 enantiomeric excess.

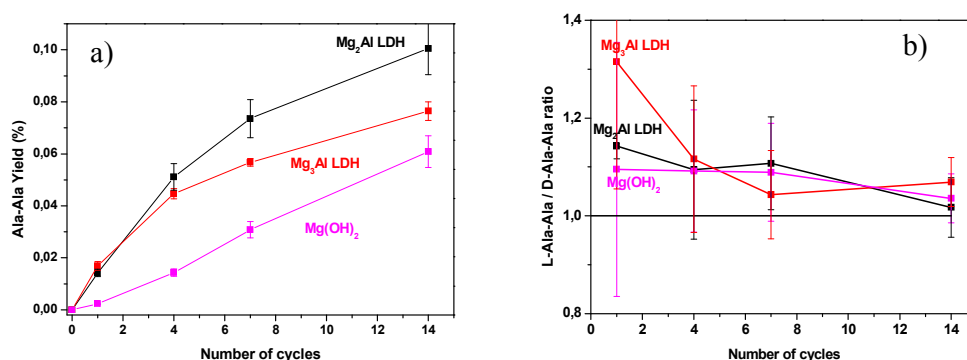
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3 364 Influence of the layer charge density LDHs with different Mg:Al ratios were also prepared,  
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5 365 along with Mg(OH)<sub>2</sub>, and used to induce peptide formation. The experimental conditions were  
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7 366 similar to those previously described. The results of these experiments are presented in  
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9 367 **Figure 4a**. In each case a continuous increase in peptide yield is observed over the number of  
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11 368 cycles, but the quantity of peptides formed depends on the layer charge density. After 14  
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13 369 cycles, the di-alanine yield reaches 0.1% for Mg<sub>2</sub>Al LDH, 0.08% for Mg<sub>3</sub>Al LDH and 0.06%  
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15 370 for Mg(OH)<sub>2</sub>. These three materials exhibit higher peptide conversion yields than those with  
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17 371 the other cations (Zn or Ni), pointing towards the potential role of the Mg materials,  
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19 372 especially given their abundance at alkaline vents.

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22 373 Compared with brucite, the presence of Al, with stronger electronegativity, slightly  
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24 374 weakens the basicity of the surface sites which seems to have a positive influence on the  
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26 375 reactivity. These results were not expected given the reasoning given above that increasing  
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28 376 the basicity will promote peptides formation. This apparent discrepancy suggests that the  
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30 377 strength of the basic sites does not provide a complete rationalization for interpreting the  
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32 378 actual yield of peptide formation when brucite and LDH materials are compared. Moreover,  
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34 379 as observed in table S1, the apparent domain size of brucite and LDH materials are largely  
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36 380 different, so that the number of active sites for peptides reaction is believed to be different.  
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38 381 As far as Mg<sub>3</sub>Al LDH is concerned and assuming an ordered distribution of the cations within  
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40 382 the layer, OH groups can be linked to three magnesium (25%) or two magnesium and one  
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42 383 aluminum (75%)<sup>66</sup>. As observed in figure 3a, the observed peptide yields is about 80% of the  
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44 384 one of Mg<sub>2</sub>Al LDH potentially indicating that the surface active sites may be Mg<sub>2</sub>Al(OH)  
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46 385 moieties. Peptide bond formation involves nucleophilic attack of the amino group on the  
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48 386 electrophilic carboxylate carbon. At pH 9-9.5, a significant population of the amino groups  
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50 387 will be protonated, which decreases its nucleophilic character. Given that the point of zero  
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52 388 charge LDH lies generally in the range [10.5 - 11.5]<sup>67-70</sup>, the LDH surface acquires a positive

charge at pH 9-9.5 that may be balanced by an increased concentration of  $\text{OH}^-$  near the surface of the particle. Therefore, the local pH at the surface of the particles may be much higher than the one in solution, enhancing the deprotonation of the amino group and promoting its nucleophilicity thus leading to a higher conversion yield.

**Figure 4b** shows the dipeptide ratio L-Ala-Ala/D-Ala-Ala over the number of cycles. Interestingly, slight enantioselectivity is observed with the L form being favoured, especially during the first cycle. With increasing number of cycles this enantiomeric excess disappears and could reflect different reactivity between L form and D form as reported for the SIPF reaction, or different surface adsorption properties.



**Figure 4.** Results from peptide forming reactions on brucite and binary Mg/Al layered double hydroxides, where  $\text{Mg}^{2+}/\text{Al}^{3+}$  was either 2 or 3, showing: (a) yields of di-alanine against the number of wetting/drying cycles; (b) the enantiomeric excess.

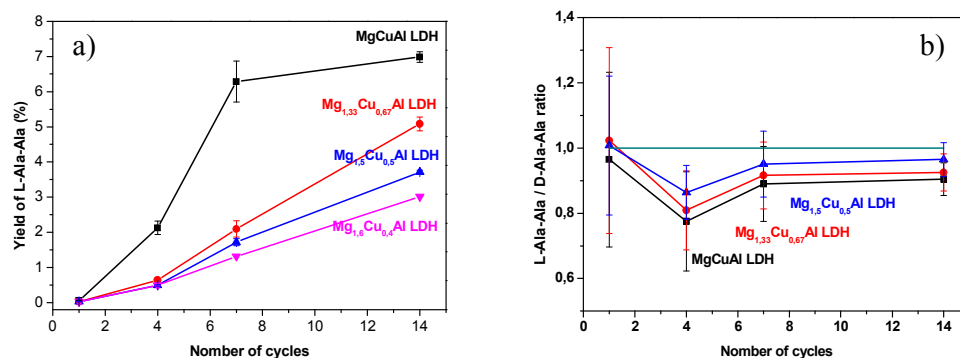
### 3.4 Assessing the role of copper in mineral induced peptide formation experiments

Mg-based layered double hydroxides are shown above to be effective substrates for the formation of alanine peptides, even in the absence of Cu(II). However, compared with the homogeneous SIPF reaction, run as a comparison, the peptide yields were much lower, and therefore aqueous Cu(II) in the SIPF reaction is a better catalyst than either the LDHs or

1  
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3 408 brucite. To investigate whether Cu(II) can play a role in LDH –catalysed peptide synthesis,  
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5 409 two sets of experiments were carried out.

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7 410 Binary LDHs with soluble copper and SIPF reagents In the first set of experiments, the SIPF  
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9 411 reaction and the LDH catalyst were combined and the simple SIPF reaction with the SIPF  
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11 412 reactants (NaCl, Cu<sup>2+</sup>, and Alanine) was carried out in the presence of Mg<sub>2</sub>Al LDH. Under  
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13 413 these conditions, no peptide was formed and it seems the LDH mineral surface inhibits the  
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15 414 SIPF reaction. It is speculated that this is due to competition between the complexation  
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17 415 reaction with the copper that is involved in the SIPF reaction and the surface adsorption sites  
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19 416 on the minerals, thus preventing any SIPF reaction.

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22 417 Ternary LDHs with structural copper In a second set of experiments, copper was instead  
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24 418 directly incorporated into the LDH layers at various concentrations to determine if structural  
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26 419 copper promotes the formation of peptides. The results, given in **Figure 5a**, show that the  
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28 420 presence of structural Cu(II) in the LDH significantly increases the conversion yield from  
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30 421 0.1% in the binary system to values that range from 2.5% - 7% in the ternary system, a  
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32 422 remarkable increase. The profiles of the curves are similar in all these phases and show a  
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34 423 continued increase in the conversion to peptides over the number of cycles. Moreover, a linear  
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36 424 correlation is observed between the quantity of Di-Ala formed after 14 days and the initial  
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38 425 quantity of copper in the layer (**Figure ESI 3**). Unlike binary LDHs, the peptide formation  
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40 426 reaction in the ternary system exhibits slight enantioselectivity toward the D-amino acids, the  
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42 427 extent of which is correlated with the quantity of copper within the layer (**Figure 5b**). The  
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44 428 reason for this remains unclear.  
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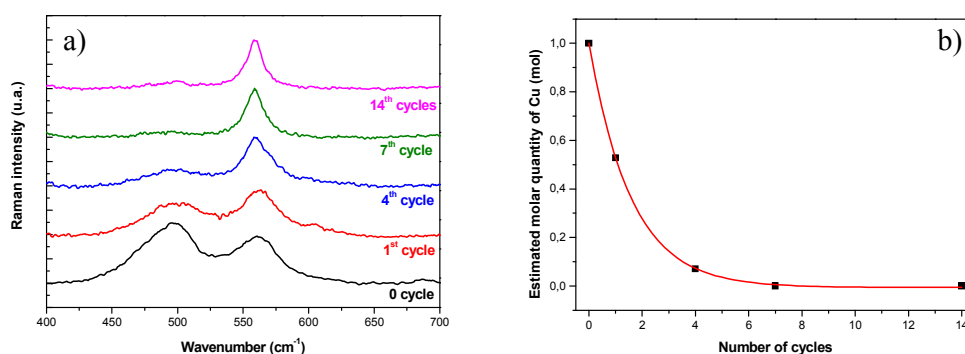
430 **Figure 5.** Results from peptide forming reactions on ternary Mg/Cu/Al layered double hydroxide minerals,  
 431 with varying Cu<sup>2+</sup> content, showing: (a) yields of di-alanine against the number of wetting/drying cycles for each  
 432 mineral; (b) the enantiomeric excess.

433

434 During the experiments, changes in the colour of the solution were observed, from a pale  
 435 green of the ternary LDH system to a dark blue. This strongly suggests that copper was  
 436 released into solution during the experiments. However, the combination of LDH and SIPF  
 437 reagents in the initial experiment yielded no dipeptide, and also the higher yields than found  
 438 in the SIPF reaction alone provide evidence that it is not the leached copper that induces the  
 439 observed reactivity.

440 In order to gain more insight into this transformation, samples were taken at different  
 441 stages of reaction and analysed by Raman spectroscopy. To ensure that the samples were  
 442 homogeneous and that the results discussed relevant, spectra were recorded at 2 μm resolution  
 443 on several different areas of each of the samples. Some differences were observed that could  
 444 possibly be assigned to other phases but the band intensities and frequencies assigned to the  
 445 LDH were seen to be highly reproducible wherever the signal were collected. It is clear from  
 446 **Figure 6** that transformation of the LDH phase occurs with increasing number of cycles as  
 447 shown by the decrease in intensity of the LDH bands at 500 cm<sup>-1</sup>, while the band located at  
 448 560 cm<sup>-1</sup> remains unchanged. The former band is assigned to Cu<sub>2</sub>Al-OH groups while the

latter is attributed to  $\text{Mg}_2\text{Al-OH}$  groups in the ternary  $\text{MgCuAl}$  system. This spectral evolution may be correlated with the partial dissolution of the  $\text{Cu}_2\text{Al}$  domain. This interpretation is further supported by the visual resemblance between the LDH taken at the 14th cycles and  $\text{Mg}_2\text{Al}$  binary LDH phases and X-ray evidence below.

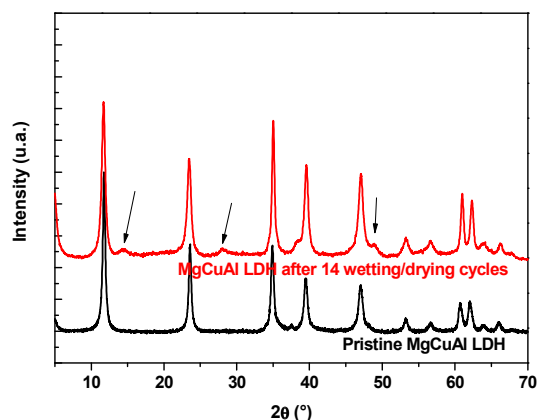


**Figure 6.** Change in the Raman spectra of  $\text{MgCuAl}$  LDH where  $\text{Mg/Cu} = 1$  showing: (a) the peaks at  $500\text{ cm}^{-1}$  and  $560\text{ cm}^{-1}$  as a function of number of wetting cycles; (b) calculated residual LDH framework  $\text{Cu}^{2+}$ .

As observed in **Figure ESI 3**, a linear correlation was found between the quantity of copper within the LDH phase and the relative intensity of the Raman band. Through applying this correlation to the data from our samples, shown in **Figure 6a**, the quantity of structural LDH copper can be estimated. As can be seen from **Figure 6b**, the quantity of copper decreases exponentially over the number of cycles, to approach 0 after the 7th cycle. To explore this in more detail, the  $\text{MgCuAl}$  samples that underwent 14 days of wetting/drying cycles were analysed by XRD and compared with the pristine  $\text{MgCuAl}$  LDH phase (**Figure 7**). After 14 cycles, the layered structure is conserved as indicated by the typical LDH signature. The cell parameters are  $a = 3.036\text{ \AA}$  and  $c = 22.716\text{ \AA}$ . The  $c$  parameter is very slightly lower than that of the pristine sample suggesting that the layer charge density has almost been conserved. However, a more drastic decrease of the  $a$ -parameter suggests that the composition of the layer has changed significantly. The value of the  $a$ -parameter is very close to that  $\text{Mg}_2\text{Al}$  LDH, implying that most of the copper has escaped from the structure as deduced from



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3 469 Raman analysis. This is further supported by the presence of new reflections shown by arrows  
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5 470 on **Figure 7** and which may be assigned to  $\text{Cu}^{\text{II}}$  phases. It is interesting to note that the  
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7 471 rehydration of the sample taken at the 14th cycle showed a complete resolubilisation of  $\text{Cu}^{\text{II}}$ ,  
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9 472 without any detection of deposited blue crystals at the bottom of the flask. Although the  
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11 473 precipitation of  $\text{CuCl}_2$  salt resulting from the drying process prior to XRD measurements may  
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13 474 be suggested, the absence of direct evidence and the restricted number of XRD peaks make  
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15 475 the determination of the Cu phase complicated.



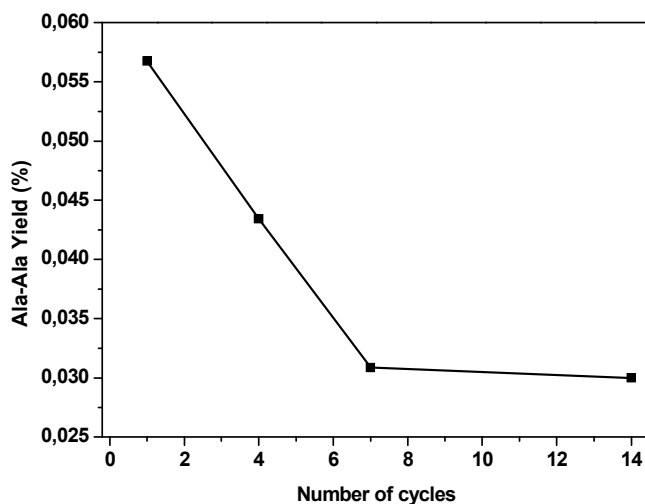
33 476  
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35 477 **Figure 7.** Powder X-ray diffraction patterns for the ternary MgCuAl layered double hydroxide before and after  
36  
37 478 14 wetting/drying cycles. Emergence of new peaks marked with arrows in the post-treatment X-ray diffraction  
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39 479 pattern.

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42 481 **Catalyst recycling of ternary LDHs with structural copper** In order to ensure that the non-  
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44 482 LDH  $\text{Cu}^{\text{II}}$  phase was not involved in the formation of the peptides observed, new experiments  
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46 483 were performed. MgCuAl LDH was first placed in a fresh solution containing the amino acid  
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48 484 and NaCl. Once the solution was fully evaporated, the materials were rinsed thoroughly with  
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50 485 water several times. Then, a new cycle was started by addition of the amino acid/NaCl  
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52 486 solution. By comparison with the previous experiments, this corresponds to an evaluation of  
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54 487 the efficiency (reusability) of the catalyst every cycle. The results are shown in Figure 8, and

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3 488 the value of the di-alanine yield after the first cycle (0.057%) corresponds reasonably well  
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5 489 with that determined in the first cycle of the previous experiments (0.065% in **Figure 5a**). A  
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7 490 continuous decrease of the quantity of peptide then appears until the 6th cycles where a steady  
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9 491 value of 0.03% is noted. The decrease of the efficiency may possibly be attributed to the  
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11 492 mineralogical transformation of the LDH phase, decreasing the quantity of basic sites. The  
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13 493 value of 0.03% is higher but relatively close to that obtained in the absence of Cu(II) with  
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15 494 Mg<sub>2</sub>Al LDH after the first cycle (i.e. 0.015%) , strengthening our hypothesis that after Cu  
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17 495 release, the remaining solid material is Mg<sub>2</sub>Al LDH. From the 6<sup>th</sup> to the 14<sup>th</sup> cycle, the  
18  
19 496 quantity of peptides formed every cycles is identical, meaning that the binary LDH shows  
20  
21 497 very good stability over time despite its low reactivity. These experiments confirm that the  
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23 498 non-LDH Cu phase does not participate in the formation of peptides. Comparison of Figure 8  
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25 499 and Figure 5a, shows the key role of copper.

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27  
28 500 As reported above, no reaction was observed when copper and amino acid were both  
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30 501 simply in contact with LDH in the combined SIPF/binary LDH experiments, and we speculate  
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32 502 that this is the result of surface scavenging of Cu(II) from solution. In contrast, incorporation  
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34 503 of Cu(II) in the solid LDH structure leads to high yields for amino acid-peptides conversion.  
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36 504 This result is interesting since in solution with the simple SIPF reactions, high copper  
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38 505 concentration is known to inhibit the formation of di-alanine <sup>71</sup>. This suggests that the reaction  
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40 506 mechanism for di-alanine peptide formation in the presence of Cu-bearing MgAl LDHs may  
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42 507 involve specific chemisorption of alanine on the Cu-LDH surface.  
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509 **Figure 8.** Di-alanine yield as a function of number of cycles for the MgCuAl system where the material was  
510 rinsed thoroughly between each wetting cycle.

#### 511 **4. Conclusions**

512 This study set out to examine the possible role of layered precipitation minerals as catalysts  
513 for the formation of peptides in alkaline media under conditions related to early Earth  
514 hydrothermal vents. The results presented here suggest that among all brucite and binary  
515 layered double hydroxide structures tested, Mg-based materials exhibit better catalytic activity  
516 than those based on Ni or Zn. In addition, reusability tests suggest that their catalytic activity  
517 is conserved over time, so that a continuous supply of peptides may have been provided by  
518 these materials. The surface basicity may explain this reactivity of Mg-based materials  
519 through the activation of the amino group, thus strengthening its nucleophilic character.  
520 Incorporation of Cu within the brucite-like sheet enhances the reactivity more than 20 times to  
521 give dipeptide yields of up to 7%. Though this high reactivity leads to a progressive  
522 dissolution of copper, under alkaline conditions we speculate that subsequent re-precipitation  
523 of the Cu within a new LDH would maintain the cycle of di-alanine formation.

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3 524 **5. Acknowledgements**  
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5

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7  
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9  
10 527 Wiseman from the Chemistry Research Laboratory of the University of Oxford for giving  
11  
12 528 access to XRD facilities.  
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16 529 **6. Supporting informations**  
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18 530 Cell parameters for various LDH composition, linear correlation to assess the Cu content in  
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20 531 layered double hydroxides using Raman spectroscopies, and correlation between the dipeptide  
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22 532 yield and the Cu content within the LDH layer.  
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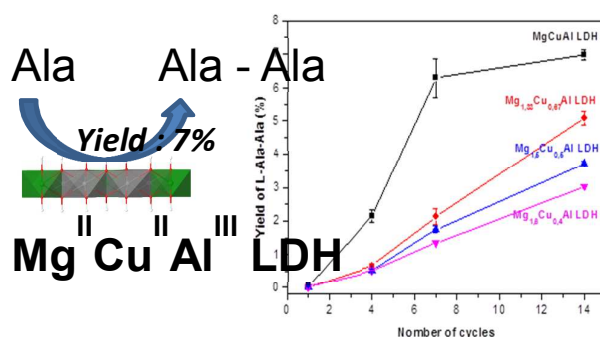
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## FOR TOC ONLY





## Peptide Formation on Layered Mineral Surfaces: The Key Role of Brucite-like Minerals on the Enhanced Formation of Alanine Dipeptides

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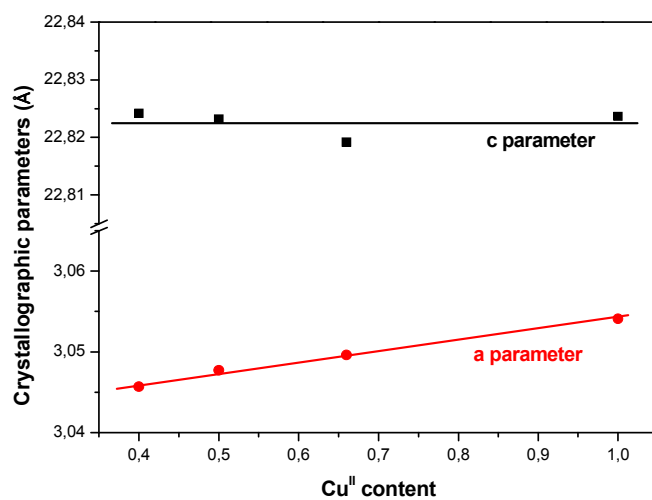
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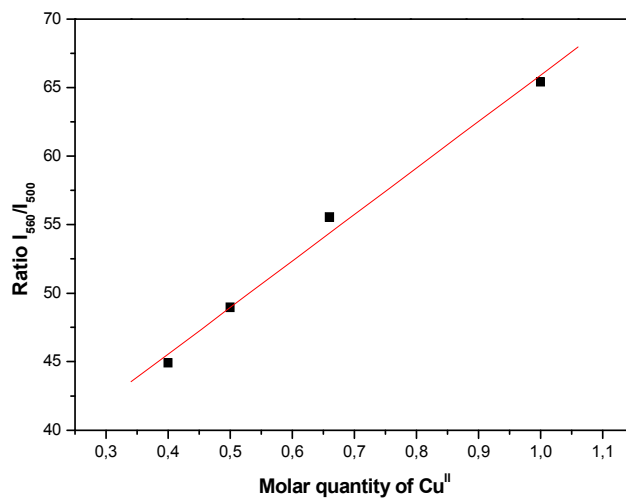
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**Table ESI 1.** Crystallographic parameters for prepared minerals used in the study.

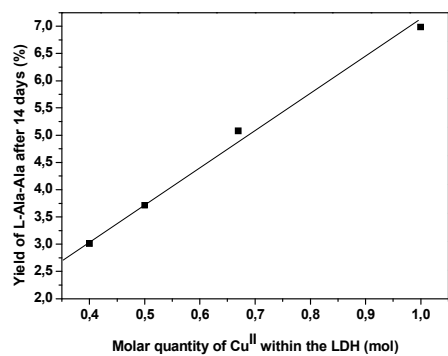
LDH	Cell parameter (Å)		Apparent domain size (nm)	
	c (Å)	a (Å)	along [001]	along [110]
Mg(OH) <sub>2</sub>	4.732	3.136	19.0	49.2
Mg <sub>2</sub> Al LDH	22.684 (7.561)	3.037	7.9	23.6
Ni <sub>2</sub> Al LDH	22.740 (7.580)	3.018	5.5	22.5
Zn <sub>2</sub> Al LDH	22.670 (7.557)	3.068	20.5	43.5
Mg <sub>3</sub> Al LDH	23.447 (7.816)	3.061	9.6	25.5



**Figure ESI 1.** Variation of the crystallographic cell parameters  $a$  and  $c$  as a function of  $\text{Cu}^{\text{II}}$  content within the layer of Ternary Mg/Cu/Al layered double hydroxides.



**Figure ESI 2.** Plotted variation in intensity of the Raman bands at  $500\text{ cm}^{-1}$  and  $560\text{ cm}^{-1}$  as a function of Cu content showing a linear correlation that can be used to assess the Cu content in Mg/Cu/Al LDH samples.



**Figure ESI 3.** Maximul yield from peptide forming reactions after 14 cycles on ternary Mg/Cu/Al layered double hydroxide minerals as a function of Cu<sup>II</sup> content of the mienral.